

## Study of the Influence of Temperature on the Deposition of SiO<sub>2</sub> Films from Reaction of Silicone Oil Vapor and Ozone Gas

Arifuzzaman Rajib<sup>1\*</sup>, Susumu Horita<sup>2</sup>, Atowar Rahman<sup>3</sup> and Abu Bakar Md. Ismail<sup>4</sup>

<sup>1</sup>*Department of Applied Physics, Electronics and Communication Engineering, Bangabandhu Sheikh Mujibur Rahman Science and Technology University, Gopalganj, Bangladesh.*

<sup>2</sup>*School of Material Science, Japan Advanced Institute of Science and Technology (JAIST), 1-1 Asahidai, Nomi, Ishikawa Prefecture 923-1211, Japan.*

<sup>3,4</sup>*Department of Applied Physics and Electronic Engineering, University of Rajshahi, Rajshahi 6205, Bangladesh.*

\*Corresponding author: *rajib.apee.38@gmail.com*

### Abstract

This work reports the influence of deposition temperature on the deposition of SiO<sub>2</sub> films on silicon substrate by using chemical reaction of silicone oil vapor and ozone gas at low temperature. An organic solution as a catalyst at atmospheric pressure has been used to enhance the deposition rate of SiO<sub>2</sub>. The deposition rate of SiO<sub>2</sub> films was found to vary with the variation of the concentration of the catalyst and deposition temperature (160°C ~ 260°C). The deposited SiO<sub>2</sub> films were confirmed by Fourier transform infrared (FT-IR) spectroscopy. The thickness and refractive index of the as-deposited films were measured by the laser ellipsometry. FT-IR spectra of the as-deposited films were very much similar to those of SiO<sub>2</sub> films found in literature. The deposition temperature was found to influence the deposition rate of SiO<sub>2</sub> strongly. The maximum deposition rate was found to 17.2 nm per minute for the case of 220°C. The deposition temperature also influenced the refractive index of the films. Experimental results showed that the deposition temperature could be a major parameter for the enhancement of the deposition rate.

*Keywords:* Low-Temperature SiO<sub>2</sub> deposition, APCVD, FT-IR spectroscopy.

### INTRODUCTION

Advanced very large scale integrated (VLSI) devices require high aspect ratio topography on semiconductor substrates. SiO<sub>2</sub> films have been widely applied in the production of electronic devices, integrated devices, optical thin film devices, sensors, and so on [1-9]. Moreover, low temperature deposition of SiO<sub>2</sub> film on a silicon substrate is desired for high quality gate oxide films to obtain high performance thin film transistors (TFTs) and to form interlayer dielectric (ILD) to suppress the disconnection of interconnect metal [10]. Recently special attention has been paid to thin-film transistor (TFT) technology because it is used for large area flat panel displays such as liquid crystal displays (LCD) [11-12]. TFT performance strongly depends on the SiO<sub>2</sub> / Si interface properties [13-15]. Conventionally, oxidation [16-17], Plasma Enhanced Chemical Vapor Deposition (PECVD) [18-21], Low Pressure CVD (LPCVD) [22-24] and Atmospheric Pressure CVD (APCVD) [25, 26] are used for deposition of SiO<sub>2</sub> on silicon substrate. In oxidation method, silicon dangling bonds are formed due to lattice mismatch between Si/SiO<sub>2</sub> interfaces which degrade the device performance [27]. LPCVD is used to deposit a wide range of possible film compositions with good conformal step coverage but an extra low

pressure system is required to attain the pressure of 10-1000 Pa (the available pressure LPCVD). PECVD method has some advantages that the advantages of a low deposition temperature as well as controlling the deposition rate quickly and accurately to obtain high-quality SiO<sub>2</sub> film [28]. However, the setup of this system is very complicated and also, sometimes the deposited films damaged due to plasma.

In this study, APCVD methods have been used to deposit SiO<sub>2</sub> films at low temperature with taking advantage of their easy set-up and lower cost (no extra setup is required for pressure control). In this method, silicone oil has been used as starting materials used to deposit SiO<sub>2</sub> films. It is a promising materials because of their low cost, high thermal stability and high safety[29,30], compared with tetra-ethyl-ortho-silicate [TEOS:Si(OC<sub>2</sub>H<sub>5</sub>)<sub>4</sub>] [31, 32], a widely used as a common source materials to deposit SiO<sub>2</sub> films. TEOS is also toxic to human eyes and throats. Moreover, deposition of SiO<sub>2</sub> films in controlled ambient requires temperature in excess of 600 °C. In this work, we have been able to deposit SiO<sub>2</sub> films at relatively lower temperature by adding a more aggressive oxidant into silicone oil. In particular, we have used ozone (O<sub>3</sub>) as aggressive oxidant. In addition, an organic solution has been added to the system to enhance the deposition rate of SiO<sub>2</sub> films.

#### EXPERIMENTAL

Figure 1 shows a schematic diagram of the atmospheric pressure chemical vapor deposition (AP) CVD system used for the deposition of silicon oxide films. Here, the conventional APCVD system has been slightly modified by introducing a new line part which facilitates to add anorganic solution to the main stream of the system.

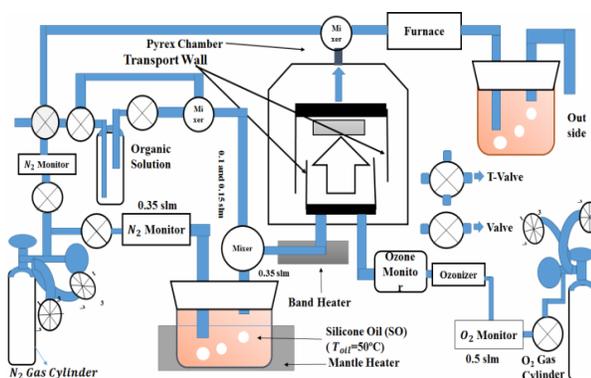
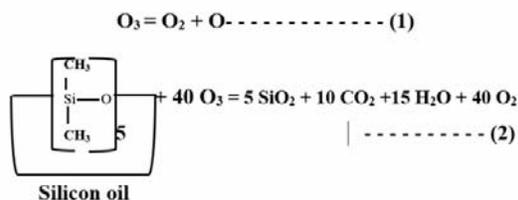


Fig 1: Schematic diagram of APCVD method.

The chemically cleaned substrates were loaded into the chamber, which was uniformly heated by heater at a deposition temperature. The silicone oil (SO) (TSF-405) was heated to about 50 °C and vaporized directly by bubbling with N<sub>2</sub> gas at a flow rate of 0.35 standard liter per minute (slm) through a stainless tube heated to about 55 °C to avoid the condensation of SO vapor. The organic solution was heated to three different temperatures of 21.9 °C, 28.1 °C and 34.1 °C by another heater and vaporized by bubbling with N<sub>2</sub> gas at five different flow rates of 0, 0.05, 0.1, 0.15 and 0.2 slm. After that, the

organic solution was mixed with silicon oil vapor and then flown into the chamber. The ozone was generated by an ozonizer from O<sub>2</sub> gas at the flow rate of 0.5 slm and then introduced into the chamber together with the silicone oil and organic solution vapor. The substrates were n-type single crystals silicon with a resistivity of 5-15 Ω-cm. The crystallographic orientation of the substrate was [111]. The dopant of the wafer is phosphorus. The thickness of the substrate = 525 ± 25 μm. They were chemically cleaned by hot ammoniacal solution (NH<sub>4</sub>OH : H<sub>2</sub>O<sub>2</sub> : H<sub>2</sub>O = 1 : 1 : 4) and hot acid solution (HCl : H<sub>2</sub>O<sub>2</sub> : H<sub>2</sub>O = 1 : 1 : 4). After each chemical solution treatment, the chemical oxide was removed by 1% HF solution.

In our experimental process, O<sub>3</sub> is decomposed to O<sub>2</sub> and reactive O shown in equation (1). This decomposition is dependent on deposition temperature [33, 34]. Oliver R. Wulf & Richard C. Tolman specified the range of decomposition temperature that is 148 °C to 179 °C [34]. Since the O is very chemically active, the overall reaction of the process of reactive oxygen and SO is shown in equation (2). In the first stage, the chemically reactive oxygen atoms are reacted with the methyl (-CH<sub>3</sub>) bond of the SO in the gas phase and broken the methyl bond to form hydroxyl (-OH) bond then intermediate products of deca-hydroxyl-cyclo-penta-siloxane (precursors) are formed together with by-products of H<sub>2</sub>O and CO<sub>2</sub>. In other words, -CH<sub>3</sub> side groups are replaced with hydroxyl (-OH) and silanol bonds of Si-OH cover the sides of the siloxane chains. In the second stage, the -OH groups on the surface are eliminated by the dehydration reaction with the -OH groups on the precursors.



Thickness and Refractive index were measured by laser ellipsometry (ULvac ESM 1A) at a wavelength 632.28 nm, spot size of about 1 nm, angle of incidence 70 °, sampling time 20 ms as well as measured 10 times at the same location. Chemical structures of as-deposited films were studied by Fourier Transform Infrared Spectroscopy (Perkin Elmer (Spectrum 100) FT-IR Spectrometer) of wave number range 500 to 4000. When the deposited films are investigated by FT-IR spectrophotometer, the deposited film is kept into the holder and waits 5 to 6 minutes for removing unwanted H<sub>2</sub>O and CO<sub>2</sub>, which are absorbed in the films.

### RESULTS AND DISCUSSION

Figure 2 shows the FT-IR spectra that were obtained from the as-deposited films with different deposition time, in the range between 500 and 4000 cm<sup>-1</sup>. Unless otherwise stated all the data was taken at ~28.1 °C and 160 ~ 260 °C for the organic solution and substrate respectively. The flow rate of N<sub>2</sub> gas was kept at 0.1 slm for the organic solution. The deposition time was also constant at 10 minutes. For the sake of simplicity, spectra are divided into two frequency regions: low frequency region between 500 and 2000 cm<sup>-1</sup> and high frequency region between 2000 and 4000 cm<sup>-1</sup>. In the low frequency

region, they consist of three broad bumps in the range between 500 and 1250  $\text{cm}^{-1}$ , showing quite different shape and width, but the comparable absorption amplitude. The line widths and the peaks position are same for all the deposition films. The broad absorption band for Si-O-Si bonds are 1010 – 1090  $\text{cm}^{-1}$  and 730 – 860  $\text{cm}^{-1}$ , which are attributed to the Si-O-Si stretching and bending vibrations, respectively [35-37]. Thus, The peaks at the wavenumbers of 1075  $\text{cm}^{-1}$  and 810  $\text{cm}^{-1}$  are labeled as Si-O(1) and Si-O(2), respectively. Another feature is observed in the range between bands 1500 to 1800  $\text{cm}^{-1}$ , which is due to the absorbed  $\text{H}_2\text{O}$  in the deposited films [35, 37, 38]. On the other hand, in high frequency region of the FT-IR spectra a broad absorption band between bands 3000 – 3600  $\text{cm}^{-1}$  are observed. A weak band at about 2300  $\text{cm}^{-1}$  is attributed as absorbed  $\text{CO}_2$  in the films [39-40].

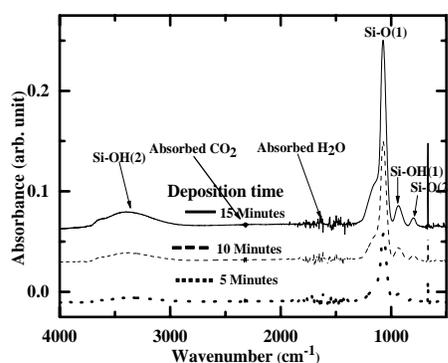


Fig 2: FT-IR spectra of silicon oxide films at temperature different deposition time.

Hydrogenated silicon oxide thin films are characterized by the absorption bands at 850 – 990  $\text{cm}^{-1}$  and 3000 – 3600  $\text{cm}^{-1}$ , which are attributed to the Si-OH stretching vibrations [40, 41]. It is observed from the figure that the amplitude of these two peaks is increased with the deposition time. The maximum peaks at the wave numbers of 950 and 3350  $\text{cm}^{-1}$  are labeled as Si-OH (1) and Si-OH(2), respectively. The FT-IR spectra showed that the films deposited from 160 to 260  $^{\circ}\text{C}$  are almost stoichiometric Si oxides without signals related to carbon, although small signals of Si-OH.

The deposition time was kept constant at 10 minutes and  $\text{N}_2$  flow rate for oil was kept constant at 0.5 slm as well as  $\text{O}_2$  flow rate was 0.5 that was also constant. The catalyst concentration was  $3.05 \times 10^{-5}$  moles/ $\text{cm}^3$ . As can be seen from Figure 3, the deposition rate increases with the deposition temperature until certain temperature, and then it starts to decrease. Increase in deposition rate with the increase in deposition temperature is a common phenomenon particularly at a low deposition temperature. At lower temperature, molecular energy of the reagents increases with increasing the deposition temperature which in turn overcomes the activation energy of the reaction. Consequently the reaction rate and deposition rate increases [42-43]. But, the saturation of the deposition rate and decline nature with the higher temperature suggests that a different chemical reaction occurs in the pathway.

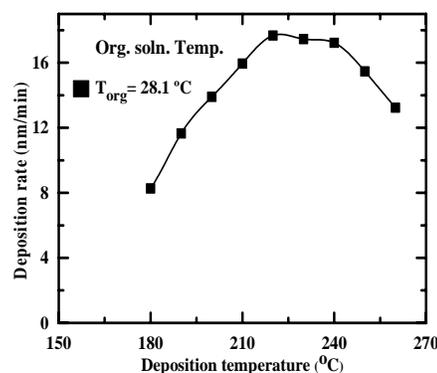


Fig 3: Deposition rate vs. deposition curve with catalyst.

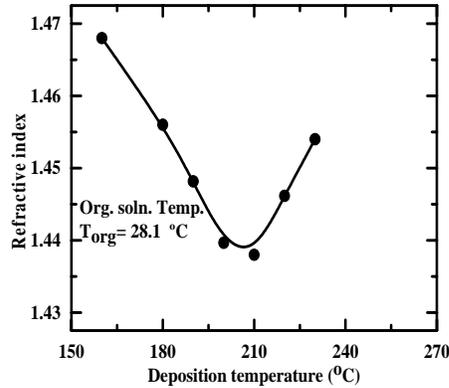


Fig. 4: Refractive index vs. deposition temperature curve at deposition time of 10 minutes.

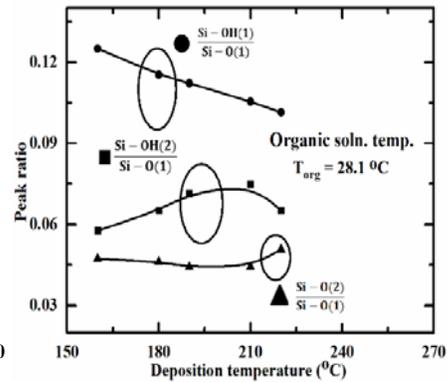


Fig. 5: Peak ratio vs. deposition temperature curve at deposition time of 10 minutes.

It can be observed from Figure 4 that the refractive index decreases with the deposition temperature and after a certain temperature the refractive index increases. The relative absorption peak intensities are plotted as a function of deposition temperature and are shown Figure 5. The peak ratios are required for qualitative discussion of refractive index.

When the amount of Si-OH(2) increases, the porosity of the film increases, because Si-OH(2) is the bond of SiO<sub>2</sub> and water. As a result decrease in density as porosity is increased in the structure. Now, from the Lorentz-Lorentz model [44-47], there is a physical relation between density and refractive index (the refractive index is directly proportional to the density of the film), So that, refractive index is inversely proportional to the Si-OH(2) bond.

It can be seen from Figure 5 that the ratio of Si-OH(1)/Si-O(1) decreases with the deposition temperature. As mentioned in experimental, -Si-O-Si- is produced from precursors. Dehydration reaction introduces production of -O-Si-O- from the precursors. It is known that, Dehydration is very active reaction. At the low deposition temperature, some of precursors are not reacted. But, when the deposition temperature increases the dehydration rate also increased, more precursors are reacted with their OH bond. So the ratio of Si-OH(1)/Si-O(1) decreases with deposition temperature.

### CONCLUSIONS

The SiO<sub>2</sub> films were deposited on a silicon substrate at atmospheric pressure and temperature between 160 °C and 260 °C by using the chemical reaction between silicone oil vapor and ozone gas as well as an organic solution was used as a catalyst. The deposition rate was too low, e.g., 1 nm/min at 200 °C for the case of without organic solution [33]. But in our study, the deposition rate was 11.85 nm/min at 200 °C by using organic solution. The average deposition rate was found to be 13.2 whereas the maximum deposition rate was 17.2 nm per minutes at 220 °C deposition temperature. The FT-IR spectra showed that the films deposited from 160 °C to 260 °C were almost

stoichiometric Si oxides. It is confirmed from FT-IR spectra of the as-deposited films. Although weak peaks from Si-OH related bond was found in the FT-IR spectra no peak related to Carbon was observed. The presence of Si-OH bond indicated the presence of small amount of pores in the deposited film that was responsible for the variation in refractive index of the deposited films. The amount of Si-OH were optimized with the help of tuning the deposition temperature that could be occurred due to increase dehydration reaction. From the experimental data it could be concluded that the deposition rate of SiO<sub>2</sub> from the chemical reaction between silicone oil vapor and ozone gas could be enhanced by optimizing the deposition temperature.

#### Acknowledgment

The work was partially funded by the special visiting student programme of JAIST through a contract with the University of Rajshahi.

#### References

- [1] N. Boumaiza, S. Achour, M.E. Tayar, "Thin SiO<sub>2</sub> growth by combined rapid thermal and plasma processing", *thin solid film journal*, **261** 1-2 (1995) 352.
- [2] H.Qu, "CMOS MEMS Fabrication Technologies and Devices", *J.Micromachines* **7** (2016) 14.
- [3] K. K. Fu, Z. Wang, J. Dai, M. Carter, and L. Hu, "Transient Electronics: Materials and Devices" *J. Chem. Mat.*, **28** (11)(2016) 3527.
- [4] G. Barillaro, A. Diligenti, A.Nannini, G. Pennelli, "A thick silicon dioxide fabrication process based on electrochemical trenching of silicon", *Sensors and Actuators* **107**, (2003) 279.
- [5] L. Xuan, G. Jie, Y. Yi, S. Yixu and R.Tianling, "Feature Scale Simulation of PECVD of SiO<sub>2</sub> in SiH<sub>4</sub>/N<sub>2</sub>O Mixture", *J. Plasma Science and Technology*, **16**(2013) 4.
- [6] K. Pfeiffer, S. Shestaeva, A. Bingel, P. Munzert, L. Ghazaryan, C. V.Helvoirt, W. M. M. Kessels, U. T. Sanli, C.Grévent, G.Schütz, M.Putkonen, I. Buchanan, D.Ristau, A.Tünnermann, and A.Szeghalmi, "Comparative study of ALD SiO<sub>2</sub> thin films for optical applications", *J.Optical Material Express*, **6**(2), (2016), 660.
- [7] S. C. D. Zhou, L. Huang, J. R. Gundlach, D. J. Jackson, T. N. Kane, M. G. Hill, I. G. Hammond, M. S. Campi, J. Greening, B. K.Franci, J. and W. John L., "Organic thin-film transistor-driven polymer-dispersed liquid crystal displays on flexible polymeric substrates", *Appl. Phys.Lett.*, **80**(6), (2002) 1088.
- [8] J. Goldberger, A. I. Hochbaum, R. Fan and P. Yang, "Silicon vertically integrated nanowire field effect transistors", *Nano Lett.*, **6** (5), (2006) 973.
- [9] A.C.R. Grayson, R.S. Shawgo, A.M. Johnson, N.T. Flynn, Y. Li ; M.J. Cima, R. Langer, "A BioMEMS review: MEMS technology for physiologically integrated devices", in the proceeding of IEEE, **92**(1), (2004) 6.
- [10] S. Higashi, D. Abe, S. Inoue, and T. shimoda: Jpn. "Low-Temperature Deposition of Silicon Oxide Film from the Reaction of Silicone Oil Vapor and Ozone Gas", *J. Appl. Phys.*, **40**, (2001) 4171.
- [11] C.Hilsum, "Flat-panel electronic displays: a triumph of physics, chemistry and engineering", *PhilosTrans A Math PhysEng Sci.* 368(1914), (2010) 1027.
- [12] Y. S. Yang, I. K. You, H. Han, J. B. Koo, S. C. Lim, S. W. Jung, B. S. Na, H. M. Kim, M. Kim and S. H. Moon, "Electrical characteristics of InZnO<sub>3</sub> transistors fabricated by reverse offset printing", *J. Ceram. Proc. Res.* **16** 2, (2015) 276.

- [13] J. Kim, T. J. Song, J. H. Kim, S. P. Cho, M. S. Yang, I. B. Kang, Y. K. Hwang, I. J. Chung, *ELSEVIER*, **31**(2), (2010) 82.
- [14] S. Imai, S. Mizushima, Asuha, Hikaru Kobayashi, "Properties of thick SiO<sub>2</sub>/Si structure formed at 120 °C by use of two-step nitric acid oxidation method", *J. of Appl. Surf. Sci.*, **254**, (2008) 8054.
- [15] C. T. Angelis, C. A. Dimitriadis, M. Miyasaka, F. V. Farmakis, G. Kamarinos, J. Brini and J. Stoemenos, "Effect of excimer laser annealing on the structural and electrical properties of polycrystalline silicon thin-film transistors", *J. Appl. Phys.*, **86**, (1999) 4600.
- [16] Sang Chul Lima, Seong Hyun Kima, Jung Hun Leea, Mi Kyung Kima, Do Jin Kimb, Taehyoung Zyunga, *Journal of Synthetic Metal*, **148**, (2005) 75.
- [17] B. E. Deal, M. Sklar, A. S. Grove and E. H. Snow, "Characteristics of the Surface Charge (Q<sub>ss</sub>) of Thermally Oxidized Silicon", *J. Elect. Chem. Soc.*, **114**(3), (1967) 266.
- [18] J. Batey and E. Tierney, "Low temperature deposition of high quality silicon dioxide by plasma enhanced chemical vapor deposition", *J. Appl. Phys.*, **60**, (1986) 3136.
- [19] C. E. Viana, A. N. R. d. Silva, N. I. Morimoto, and O. Bonnaud, "Analysis of SiO<sub>2</sub> Thin Films Deposited by PECVD Using an Oxygen-TEOS-Argon Mixture", *Braz. J. Phys.*, **31**(2), (2001).
- [20] M. R. Amirzada, A. Tatzel, V. Viereck, H. Hillmer, "Analysis of SiO<sub>2</sub> Thin Films Deposited by PECVD Using an Oxygen-TEOS-Argon Mixture Surface roughness analysis of SiO<sub>2</sub> for PECVD, PVD and IBD on different substrates", *Appl. Nanoscience*, **6**(2), (2016) 215.
- [21] K. H. A. Bogart, N. F. Dalleska, G. R. Bogart, and E. R. Fisher, "Plasma enhanced chemical vapor deposition of SiO<sub>2</sub> using novel alkoxysilane precursors", *J. Vac. Sci. & Technol.* **13**(2), (1995) 476.
- [22] R. Weber, G. Wahl, R. Huber, "LTO - SiO<sub>2</sub> Deposition in a Stagation Flow LPCVD System", *J. de Phys. Colloq.*, **50**, (1989) 411.
- [23] K.E. Goodson ; M.I. Flik ; L.T. Su ; D.A. Antoniadis, "Annealing-temperature dependence of the thermal conductivity of LPCVD silicon-dioxide layers", *IEEE Electron Device Letters*, **14**, (1993) 490.
- [24] S. E. Babayan, J. Y. Jeong, V. J. Tu, J. Park, G. S. Selwyn and R. F. Hicks, "Deposition of silicon dioxide films with an atmospheric-pressure plasma jet", *Plasma Sources Sci. Technol.* **7**, (1998) 286.
- [25] Na Liu, Jia Zhang, Yunfeng Qiu, Jie Yang, Ping An Hu, "Fast growth of graphene on SiO<sub>2</sub>/Si substrates by atmospheric pressure chemical vapor deposition with floating metal catalysts", *Sci. China Chem.*, **59**(6), (2016) 707.
- [26] S.-K. Rha, T. P. Chou, G. Cao, Y.-S. Lee, and W.-J. Lee, "Characteristics of silicon oxide thin films prepared by sol electrophoretic deposition method using tetraethylorthosilicate as the precursor", *Curr. Appl. Phys.*, **9**(2), (2009) 551.
- [27] M. Okoshi, M. Kuramatsu, and N. Inoue: Jpn. "Pulsed Laser Deposition of SiO<sub>2</sub> Thin Films with Dimethylpolysiloxane Targets" *J. Appl. Phys.* **40**, (2001) 41.
- [28] H. Takao, M. Okoshi, and N. Inoue: Jpn. "Photochemical Deposition of SiO<sub>2</sub> Thin Films Using an F<sub>2</sub> Laser", *J. Appl. Phys.* **42**, (2003) 461.
- [29] Castellarin A, Grigorian R, Bhagat N, "Vitrectomy with silicone oil infusion in severe diabetic retinopathy", *Br. J. Ophthalmol.*; **87** (2003) 318.
- [30] D. McLeod, "Silicone oil in diabetic vitrectomy", *Br. J. Ophthalmol.*, **87**(10), (2003), 1303.

- [31] H. Nakashima, K. Omae, T. Takebayashi, C. Ishizuka and T. Uemura, "Toxicity of silicon compounds in semiconductor industries", *J. Occup. Health* **40**, (1998) 270.
- [32] Moretto, Hans-Heinrich; Schulze, Manfred; Wagner, Gebhard "Silicones". *Ullmann's Encyclopedia of Industrial Chemistry*. Weinheim: Wiley-VCH. doi:10.1002/14356007.a24\_057 (2005).
- [33] S. Horita, K. Toriyabe and K. Nishioka, "Low-Temperature Deposition of Silicon Oxide Film from the Reaction of Silicone Oil Vapor and Ozone Gas", *J. J. Appl. Phys.* **48**, (2009) 035502.
- [34] O. R. Wulf\* And R. C. Tolman, "The Thermal Decomposition of Ozone", *Gates Chemical Laboratory, California Institute of Technology, Pnas. Org.* **13**, 272 (1927).
- [35] H. Rinnert, M. Vergnat, "Structure and optical properties of amorphous silicon oxide thin films with different porosities", *J. Non-Crystalline Solids*, **320**, (2003) 64.
- [36] J. A Theil, D.V. Tsu, M. W. Watkins, S.S Kim, and G. Lucovsky, "Local bonding environments of SiOH groups in SiO<sub>2</sub> deposited by remote plasma-enhanced chemical vapor deposition and incorporated by postdeposition exposure to water vapor", *J. Vac. Sci. Technol.* **A8**(3)(1990) 1374.
- [37] H. Juarez, M. Pacio, T. Diaz, E. Rosendo, G. Garcia, A. Garcia, F. Mora and G. Escalante, "Low temperature deposition: properties of SiO<sub>2</sub> films from TEOS and ozone by APCVD system", *J. Phys.*, **167**, (2009) 012020.
- [38] J. Arno, Z. Yuan, and S. Murphy, "Fourier transform Infrared Characterization of Downstream Gas-Phase Species Generated by Tetraethylorthosilicate/Ozone Atmospheric pressure Reactions", *J. Elect. Chem. Soci.*, **146** (1), (1999) 276.
- [39] W. Rzodkiewicz and A. Panas, "Application of spectroscopic ellipsometry for investigations of compaction and de-compaction state in Si-SiO<sub>2</sub> systems" *Journal of Physics*, **181**, (2009) 012035.
- [40] W.A. Pliskin, "Comparison of properties of dielectric films deposited by various methods", *J. Vac. Sci. Technol.* **14**, (1977) 1064.
- [41] T. Longjuan, Z. Yinfang, Y. Jinling, L. Yan, Z. Wei, X. Jing, L. Yunfei, and Y. Fuhua, "Surface recombination velocity on a SiN<sub>x</sub>-passivated surface for PECVD silicon nitride film", *Journal of Semiconductor*, **30** (9), (2009) 096005.
- [42] G. Lucovsky, J. Yang, S.S. Chao, J.E. Tyler, W. Czubytyj, "Nitrogen-bonding environments in glow-discharge—deposited a-Si:H films", *Phys. Rev.* **29**(1984) 2303.
- [43] G. Lucovsky, J.D. Jaonnopoulos (Eds.), *The Physics of Hydrogenated Amorphous Silicon II*, Springer, Berlin, (1984) 235.
- [44] Y. B. Park and S. W. Rhee, "Microstructure and interfacial states of silicon dioxide film grown by low temperature remote plasma enhanced chemical vapor deposition", *J. Appl. Phys.* **66**, (1995) 3477.
- [45] L. Banyai, P. Gartner, "Clausius-Mosotti limit of the quantum theory of the electronic dielectric constant", *Phys. Rev.* **B 29**, (1984) 728.
- [46] Oughstun K. E., Cartwright N. A., "On the Lorentz-Lorenz formula and the Lorentz model of dielectric dispersion: addendum", *Opt. Express*, **11**, (2003) 2791.
- [47] K. Vedam, P. Limsuwan, "Piezo- and Elasto-optic Properties of Liquids under High Pressure. II: Refractive Index vs Density" *J. Chem. Phys.* **69**, (1978) 4772.