Fabrication of Epoxide Functional Hydrophobic Composite Polymer Particles by Suspension Polymerization and Subsequent Doping with Fe$_3$O$_4$ Nanoparticles

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Abstract

This investigation described a simple three-step process for the fabrication of micrometer-sized magnetic composite polymer particles. This composite polymer particle consisted of crosslinked hydrophobic poly(lauryl methacrylate-divinyl benzene) (P(LMA-DVB)) core, prepared by suspension polymerization. Then, P(LMA-DVB) copolymer core particles were coated with poly(glycidyl methacrylate) (PGMA) by seeded polymerization to introduce epoxide functionality. Finally, P(LMA-DVB)/PGMA composite particles were doped with iron oxide (Fe$_3$O$_4$) nanoparticles following in situ co-precipitation of Fe$^{2+}$ and Fe$^{3+}$ from their alkali aqueous solution. The presence of strained oxirane ring derived from PGMA segment present at the surface is expected to induce high affinity towards precipitated magnetic Fe$_3$O$_4$ nanoparticles. The compositional structure of P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles was confirmed by Fourier Transform IR (FTIR), electron microscopy, thermogravimetry (TG), X-ray diffraction (XRD) and energy-dispersive X-ray (EDX).

Keywords: Epoxide functionality; Hydrophobic; Doping; Fe$_3$O$_4$; Composite particles.

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1. Introduction

Micrometer-sized hydrophobic latex particles find extensive uses such as resins for chromatographic separation, water purification, oil absorbency agents, viscosity modifiers, oil-soluble drag reducers etc. [1-4]. During the last few years we have developed techniques to prepare variable-sized latex particles ranging from nano to several microns from lauryl methacrylate (LMA) a highly hydrophobic monomer because

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of its long chain alkyl group. The poor water solubility of LMA (<<0.01 g/100g at 25°C) prevents this monomer from being polymerized by conventional emulsion polymerization and the conversion is often very low [5]. In a recently published article we proposed modified emulsion copolymerization of LMA and methyl methacrylate (MMA) to produce sub-micrometer sized latex particles in a dispersion medium comprising water-ethanol mixture using water soluble initiator [6]. Ethanol was used to improve the solubility of LMA in water, a prime requirement for emulsion polymerization. The effect of third monomer on the overall emulsion terpolymerization of LMA and MMA was also investigated in terms of time-conversion graphs, morphology and size of particles [7]. Preparation of PLMA homopolymer latex particles by suspension polymerization using poly (vinyl alcohol) (PVA) as steric stabilizer was also reported [8]. The polymerization produced increased coagulation because of soft and flexible structure resulted from low glass transition temperature of PLMA [9]. To overcome this problem the author successfully carried out suspension copolymerization of LMA with high amount of divinyl benzene (DVB) as cross-linker in stable isolated droplets containing either n-hexadecane (HD) or HD–toluene (HD–T) or toluene as a non-solvent for the resulting P(LMA–DVB) copolymer [10]. In that case, the non-solvent–copolymer combinations produced variation of the internal morphology of polymer particles. In another investigation, the author proposed a novel way to prepare magnetic nanocomposite particles coated with highly cross-linked PLMA layer and evaluated their usefulness as adsorbent for removing organic pollutants from water [11]. In a relatively recent investigation the author investigated the suitability of activated swelling method for preparing several micrometer-sized polystyrene (PS)/PLMA-DVB) composite polymer particles [12]. However, this swelling method was not helpful for the preparation of monodispersed particles though the average size was in the micrometer range. These composite particles were functionalized with epoxide groups and applied for the removal of dye molecules from water solution to see the cumulative effect of functionality and hydrophobicity on adsorption behavior. In the present investigation, we used a simple suspension polymerization to prepare cross-linked P(LMA-DVB) composite particles. The prepared micrometer-sized particles were then functionalized with epoxide groups via seeded polymerization with poly (glycidyl methacrylate) (PGMA). The surface of prepared P(LMA-DVB)/PGMA composite particles were then modified to prepare magnetically doped composite particles via complexation of Fe₃O₄ nanoparticles with epoxide functionality [13]. A detail scheme for the preparation of P(LMA-DVB)/PGMA/Fe₃O₄ composite particles is shown in Fig. 1. These composite particles can find application as good adsorbent for the removal of toxic metal ions from polluted water because Fe(III)-bearing materials are known to have high selectivity and affinity for many metal ions compared to other conventional adsorbent materials like activated carbon, soil, resin and alumina [14-16]. The highly reactive epoxide functionality can also be used either directly or through modification to immobilize biopolymers, drugs, dyes and other sensitive compounds [17-21].
2. Experimental

2.1. Chemicals and instruments

LMA from Fluka Chemika (Switzerland) was repeatedly rinsed with 10% NaOH aqueous solution to remove any inhibitor and then passed through activated basic alumina in a column chromatography. Crosslinking agent DVB (80% grade) from Sigma-Aldrich, Chemie (USA) was purified with aqueous 10% NaOH solution and subsequently dehydrated by stirring with anhydrous CaCl$_2$. Benzyol peroxide (BPO) from BDH Chemicals Ltd. (UK) was recrystallized from methanol and preserved in the refrigerator before use. Poly(vinyl alcohol) (PVA) from Thomas Baker Chemicals Limited (India) of molecular weight $1.4 \times 10^3$ gmol$^{-1}$ was used as polymeric stabilizer. Ferric chloride hexahydrate (FeCl$_3$.6H$_2$O), ferrous sulfate (FeSO$_4$), NH$_4$OH, oleic acid and other chemicals were of analytical grade. Deionized water was distilled using a glass (Pyrex) distillation apparatus.

Scanning electron microscopy, SEM was performed to see the particle size distribution with a SU8000 microscope (Hitachi, Japan) operating at a voltage of 20 kV. Energy dispersive X-ray (EDX) spectral analyses were also performed (SU8000) to confirm the elemental composition. FTIR (Perkin Elmer, FTIR-100, USA); X-ray diffractometer (Rigaku Ultima IV RINT D/max-kA, Japan), and thermogravimetry with a TGA EXSTAR-6000 (Seiko Instrument Inc., Japan) were used for the characterization of the latex particles.

2.2. Preparation of crosslinked P(LMA-DVB) core particles

3 g of each LMA and DVB, 1.2 g of PVA and 0.12 g of BPO were taken in a 100 mL stopper bottle. 20 mL distilled water was added and the mixture was magnetically stirred at 2000 rpm for 1 h at 20°C to form microdroplets. In order to emulsify completely the monomer mixture was further homogenized by a homogenizer (Digital Ultra Turrax, IKA, Germany) at 20,000 rpm for 3 min in ice-water bath. The homogenized mixture was transferred to a three-necked round flask immersed in a thermostat water bath kept at 75°C and make up water was immediately added to adjust the total volume to 200 mL.
The polymerization was continued for 24 h under a nitrogen atmosphere while the reaction mixture was mechanically stirred at 100 rpm. The P(LMA-DVB) copolymer particles were washed repeatedly with double distilled water.

2.3. Preparation of crosslinked \( P(\text{LMA-DVB})/\text{PGMA} \) composite polymer particles

\( P(\text{LMA-DVB})/\text{PGMA} \) composite polymer particles were prepared by seeded polymerization of 1.5 g of GMA in presence of 3.0 g of P(LMA-DVB) seed particles employing 0.03 g of V-50 as initiator. Distilled water (150 g) was used as dispersion medium. The polymerization was carried out in a three-necked round bottomed flask set at 70°C in a thermostat water bath for 12 h in a nitrogen atmosphere. \( P(\text{LMA-DVB})/\text{PGMA} \) composite polymer particles were washed by replacing the continuous phase with double distilled water following centrifugation prior to characterization.

2.4. Preparation of crosslinked \( P(\text{LMA-DVB})/\text{PGMA}/\text{Fe}_3\text{O}_4 \) composite polymer particles

The surface of prepared \( P(\text{LMA-DVB})/\text{PGMA} \) composite particles were modified to prepare magnetically doped \( P(\text{LMA-DVB})/\text{PGMA}/\text{Fe}_3\text{O}_4 \) composite particles. The coprecipitation of \( \text{Fe}^{2+} \) (0.556 g) and \( \text{Fe}^{3+} \) (0.6255 g) from their alkali aqueous solution (molar ratio 1:2) was carried out in presence of 2.5 g of \( P(\text{LMA-DVB})/\text{PGMA} \) composite polymer particles dispersed in 150 g of water containing 20 g of 25% \( \text{NH}_3\text{OH} \) and. The precipitation was carried out in a three-necked round bottom flask under a nitrogen atmosphere at 60°C for 2 h. Towards the end of precipitation process oleic acid (0.019 g) was added as stabilizer and the mixture was stirred for another 30 min. The composite emulsion was black in color, washed repeatedly by magnetic separation and decantation.

2.5. SEM observation

A drop of latex dispersion was taken on a brush sample holder and dried under vacuum at room temperature. The dried sample was coated with 2 Å thickness gold layer before observation under SEM.

2.6. Thermogravimetric analysis (TGA)

Thermal properties of the washed and dried powder were measured by heating samples under flowing nitrogen atmosphere from 30°C to 700°C at a heating rate of 20°C/min and the weight loss was recorded.

2.7. X-ray diffraction (XRD)

The vacuum dried powder samples were placed on a glass sample holder. The XRD patterns of the powder samples were recorded by scanning X-ray diffractometer using Cu
Kα radiation at room temperature. The intensities were measured at diffracted angle, (2-theta) from 10° to 90° at a continuous scan rate of 1°/min.

3. Results and Discussion

Fig. 2 shows the SEM photographs of P(LMA-DVB) seed particles, P(LMA-DVB)/PGMA and P(LMA-DVB)/PGMA/Fe₃O₄ composite polymer particles. The average size and coefficient of variations (CVs) are 9.44 µm and 54.88 % for P(LMA-DVB) seed particles, 8.55 µm and 20.31 % for P(LMA-DVB)/PGMA and 11.26 µm and 45.78 % for P(LMA-DVB)/PGMA/Fe₃O₄ composite particles respectively. It is evident that seed particles have broad particle size distribution as normally observed for latex particles prepared by suspension polymerization [22-25]. Due to very high polydispersity index it is not wise to follow the increase in average particle size after seeded polymerization. The surface of P(LMA-DVB) seed particles is relatively smooth. Following seeded polymerization with GMA the surface of P(LMA-DVB)/PGMA composite particles becomes rough with multiple protrusions. This normally happens when phase separation occurs during swelling of highly crosslinked polymer seed particles with monomer [12,26,27]. The surface heterogeneity increases further due to precipitation of iron oxide nanoparticles on the surface of P(LMA-DVB)/PGMA composite polymer particles. The interaction of epoxide ring on the surface of P(LMA-DVB)/PGMA composite polymer particles was followed by coordination with Fe₃O₄ nanoparticles [13]. The formation of hydrogen bond between hydroxyl groups on the surface of iron oxide and ester groups of PGMA can also induce strong fixation of nanoparticles [28,29].

Fig. 2. SEM photographs of a) P(LMA-DVB) seed particles, b) P(LMA-DVB)/PGMA and c) P(LMA-DVB)/PGMA/Fe₃O₄ composite polymer particles. Fig. inset (c) shows magnified particle image.
FTIR is a useful tool to identify the surface components of polymer particles. Fig. 3 shows the FTIR spectra of P(LMA-DVB) seed particles, P(LMA-DVB)/PGMA and P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles. In the spectrum of P(LMA-DVB) seed particles the characteristic sharp stretching vibration of ester carbonyl group derived from LMA appears at 1710 cm$^{-1}$. The absorption signals in the range 2800-3000 cm$^{-1}$ correspond to aliphatic and aromatic –CH stretching vibrations. Following seeded polymerization with GMA two characteristic absorption bands due to epoxide group appear at 991 and 915 cm$^{-1}$ in P(LMA-DVB)/PGMA composite polymer particles. Comparatively in the spectrum of P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles the absorption bands at 551 and 465 cm$^{-1}$ represent the characteristic stretching vibrations of Fe-O bonds [30,31]. The intensity of the characteristic epoxide signals reduce following co-ordination of Fe$_3$O$_4$ nanoparticles. The shape of the broad signal averages at 3485 cm$^{-1}$ changes following the precipitation of Fe$_3$O$_4$ nanoparticles on P(LMA-DVB)/PGMA composite polymer particles. This signal attributes to the stretching vibration of surface water molecules and non-dissociated –OH groups derived from oleic acid or the envelope of hydrogen bonded surface –OH groups. The above results confirm the formation of epoxide functional composite polymer particles and subsequent doping with Fe$_3$O$_4$ nanoparticles.

Fig. 3. FTIR spectra of a) P(LMA-DVB) seed particles, b) P(LMA-DVB)/PGMA and c) P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles.

Fig. 4 illustrates the comparative XRD diffraction patterns of P(LMA-DVB) seed particles, P(LMA-DVB)/PGMA and P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles. Both P(LMA-DVB) seed particles and P(LMA-DVB)/PGMA composite polymer particles exhibits broad and diffused characteristic amorphous signal at two theta value of 15°. In contrast the XRD pattern of magnetically doped P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles exhibits six characteristic sharp signals
for iron oxide nanoparticles at 30.3°, 35.6°, 43.5°, 54.0°, 56.6° and 62.46° which can be assigned to (220), (311), (400), (422), (511) and (440) respectively according to the database of magnetite cited in JCPDS-International Centre (JCPDS Card: 19-0629) [32]. This XRD pattern of magnetically doped composite polymer particles additionally shows the broad diffuse amorphous signal for polymer segment as well. This result confirms the doping of P(LMA-DVB)/PGMA composite polymer particles with Fe₃O₄ nanoparticles.

Fig. 4. XRD patterns of a) P(LMA-DVB) seed particles, b) P(LMA-DVB)/PGMA and c) P(LMA-DVB)/PGMA/Fe₃O₄ composite polymer particles.

Fig. 5 shows mass loss against temperature from TG analyses measured for P(LMA-DVB) seed particles, P(LMA-DVB)/PGMA and P(LMA-DVB)/PGMA/Fe₃O₄ composite polymer particles. As the temperature increases to 700°C the organic polymer in both P(LMA-DVB) seed particles and P(LMA-DVB)/PGMA composite polymer completely burned away leaving no residual mass. Comparatively in magnetically doped P(LMA-DVB)/PGMA/Fe₃O₄ composite polymer particles a residual mass of ca. 16% is left, which is assumed to be the percentage of iron oxide incorporated in composite particles. Irrespective of composition of latex particles, TGA thermograms exhibit two distinct mass loss regions. The initial mass loss in the region 200° to 400°C is associated with the loss of associated moisture, degradation of oligomer and linear polymer chain. The second mass loss region starts at higher temperature (450°C) corresponds to thermal degradation of crosslinked polymer. Overall the mass loss onset temperature in P(LMA-DVB)/PGMA composite polymer particles reduces to lower temperature possibly due to the decrease in crosslinking density following modification with PGMA layer. The doping of P(LMA-DVB)/PGMA composite polymer particles with inorganic Fe₃O₄ nanoparticles again increases the mass loss onset temperature and almost overlaps with crosslinked P(LMA-DVB) seed particles. These results suggest that the thermal stability of magnetically doped epoxide functional composite particles improved following the incorporation of inorganic Fe₃O₄ nanoparticles.
Fig. 5. TGA thermograms of a) P(LMA-DVB) seed particles, b) P(LMA-DVB)/PGMA and c) P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles.

A comparative EDX plot of P(LMA-DVB) seed particles, P(LMA-DVB)/PGMA and P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles is shown in Fig. 6. The EDX spectra of P(LMA-DVB) seed and P(LMA-DVB)/PGMA composite polymer particles show signal due to C and O elements. However, the amount of O (atom %) increases in P(LMA-DVB)/PGMA composite polymer particles from 6.6% to 21.9% after seeded polymerization with epoxide functional GMA monomer. The EDX spectrum of P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles indicates the presence of additional signal due to Fe (2.54 atom %) and the amount of O (atom %) further increases to 23.49%. These results confirm the modification of P(LMA-DVB) seed particles with epoxide functionality followed by doping with iron oxide nanoparticles.

Finally, the magnetic property of P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles was investigated. The magnetically doped composite polymer particles can be separated up to 95% by employing magnetic field within one min (Fig. 7) leaving almost transparent supernatant. As the magnetic field is removed the magnetic particles again dispersed and the process was reversible. These results suggest that the prepared P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite particles have strong magnetic property and can be separated from the treatment solution without employing time consuming separation process like centrifugation and sedimentation.
4. Conclusion

This investigation discusses the preparation of micron sized magnetically doped P(LMA-DVB)/PGMA/Fe$_3$O$_4$ composite polymer particles via a three-step process. In the first step P(LMA-DVB) core particles were prepared by suspension polymerization. The surface of core particles was modified with reactive epoxide group by seeded polymerization with GMA. The fixation of Fe$_3$O$_4$ nanoparticles with epoxide group was finally done by
Epoxide Functional Hydrophobic Composite precipitation of Fe$^{2+}$ and Fe$^{3+}$ from their alkali solution. The prepared composite polymer particles containing ca. 16% Fe$_3$O$_4$ nanoparticles can be separated by applying magnetic field. These composite polymer particles can be useful for the treatment of wastewater discharged from industry.

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