SUBSTRATE TEMPERATURE EFFECT ON THE STRUCTURAL AND OPTICAL PROPERTIES OF ZnSe THIN FILMS

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ABSTRACT

Zinc selenide (ZnSe) thin films were deposited on to chemically and ultrasonically cleaned glass substrates at different substrate temperatures from room temperature to 200° C keeping the thickness fixed at 300 nm by using thermal evaporation method in vacuum. The structural properties of the films were ascertained by X-ray diffraction (XRD) method utilizing a diffractometer. The optical properties were measured in the photon wavelength ranging between 300 and 2500 nm by using a UV-VIS-NIR spectrophotometer. The XRD patterns reveal that the films were polycrystalline in nature exhibiting f.c.c zincblende structure with average lattice parameter, a = 5.6873Å. The grain size, strain and dislocation densities of the films have been calculated. The optical transmittance and reflectance were utilized to compute the absorption coefficient, band gap energy and refractive index of the films. The band gap energy of the films was extracted from the absorption spectra. The direct band gap energy of the films slightly increases with substrate temperature.

Key words: ZnSe thin films, Substrate temperature, Structural and optical properties

INTRODUCTION

The ZnSe, a wide band gap semiconductor has high potential for application in optoelectronic devices. Because of its large band gap, ZnSe has been used as window layer for the fabrication of photovoltaic solar cells. For high efficiency solar cells, a high band gap material is required for maximum transmission of solar spectrum, which increases the open circuit voltage of the solar cell. Since solar cell window material CdS having band gap of 2.4 eV, the blue region of solar spectrum, ZnSe having band gap between 2.799 and 2.803 eV is an alternative to CdS material. Particularly, ZnSe is an interesting II-VI compound semiconducting material, widely used in optoelectronic devices, because its band gap energy belongs to the visible region (Huanyong and Wanqi 2003). Therefore, there is currently a major interest in ZnSe based materials suitable for the fabrication of light emitting devices operating in the blue-green region (Guha *et al.* 1992) and in the manufacture of optical components, mirrors, lenses etc. for IR lasers

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(Guozhen *et al.* 2003, Yi *et al.* 1994). A number of methodologies are employed in the formation of high quality ZnSe thin films, including such as chemical vapour deposition, MOCVD, electrodeposition, photochemical deposition, chemical bath deposition (CBD), pulsed laser deposition and thermal evaporation (Chaliha *et al.* 2008, Choudhury *et al.* 2004, Chu *et al.* 1992, Kale and Lokhande 2005, Kumaresan *et al.* 2002, Perna *et al.* 2002). However, thermal evaporation is extremely simple and viable compared to other cost intensive methods. Here effects of substrate temperature on the structural and optical properties of ZnSe thin films have been presented.

MATERIALS AND METHODS

The films of ZnSe were deposited on to cleaned glass substrates by thermal evaporation method in vacuum ($\approx 10^{-6}$ mbar) by using an oil diffusion pump (E306A, Edwards, UK). The source material 99.99% purity ZnSe powder (supplied by British Drug House, London, UK) was evaporated from a molybdenum boat and the substrate was placed at a distance of 10 cm. The substrate temperature of the films varied from room temperature to 200°C keeping the film thickness fixed at 300 nm (\pm 10 nm). The thickness and rate of evaporation of the films was measured *in situ* by the FTM 5 quartz crystal thickness monitor (Edwards, UK).

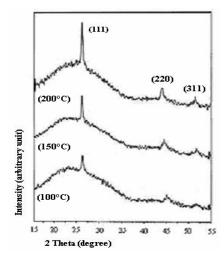
The X-ray diffraction (XRD) was used to investigate the structure of ZnSe thin films. The diffraction patterns were recorded using a Philips PW 3040 X'Pert PRO XRD system with Cu-K α radiation, operated at 40 kV and 30 mA, with angular range $15^{\circ} \le 2\theta \le 55^{\circ}$.

The thermal evaporation method was employed (Bhuiyan and Hasan 2006, 2007, Bhuiyan *et al.* 2008) by using an oil diffusion pump in vacuum to deposit ZnSe thin films. The variations of transmittance and specular absolute reflectance of the films with wavelength of light incident on them were measured using a dual beam UV-VIS-NIR recording spectrophotometer in the photon wavelength range between 300 and 2500 nm.

RESULTS AND DISCUSSION

Fig. 1 shows the X-ray diffraction spectra of three ZnSe thin films having different substrate temperature deposited at 300 nm thickness. The spectra were obtained by scanning 2θ in the range between 15 and 55° . The presence of sharp peaks confirms the polycrystalline nature of the films. The diffraction spectrum has not been found for asdeposited film. It is observed that the films prepared at higher substrate temperature are polycrystalline in nature. The peak intensity increases with increasing substrate temperature. The diffraction spectra display the characteristics diffraction peaks of the cubic phase of ZnSe. The peak at $2\theta \approx 27.26^{\circ}$ is attributed to the X-ray reflection from

the (111) planes of all the films. From the films deposited at higher substrate temperature, three prominent diffraction peaks were observed at 2θ values at 27.26, 45.18 and 53.42° corresponding to (111), (220) and (311) planes, respectively which indicates random orientation of crystallites in these films (JCPDS card No. 05-0522).



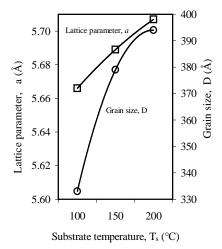


Fig. 1. X-ray diffraction spectra of ZnSe thin films having different substrate temperatures.

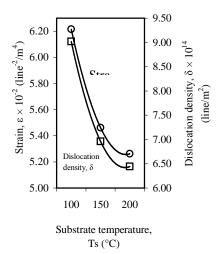
Fig. 2. Dependence of lattice parameters and grain size of ZnSe thin films having different substrate temperatures.

However, the intensities of the (220) and (311) peaks are extremely low in comparison with the (111) one. This indicates a preferential orientation of micro-crystallites with the (111) direction. The structural parameters are summarized in Table 1.

Table 1. Calculated values of structural parameters of ZnSe thin films having different substrate temperatures.

Substrate temperature T _s (°C)	Plane (hkl)	Lattice parameter $a(')$	Grain size $D^{'}$	Strain $\epsilon \times 10^{-2}$ (line ⁻² /m ⁴)	$\begin{array}{c} Dislocation \\ density \\ \delta \times 10^{14} \ (line/m^2) \end{array}$
100	111 220	5.666	333	6.215	9.018
150	111 220 311	5.689	379	5.462	6.962
200	111 220 311	5.707	394	5.262	6.442

Fig. 2 shows the variation of lattice parameter and grain size with different substrate temperatures. It is observed that both the lattice parameter and the grain size increases with the substrate temperature that confirms reasonably well to the literature (Chaliha *et al.* 2008). The concentration of lattice imperfections decreases with the increase in the substrate temperature due to the decrease in the internal micro-strain within the films and an increase in the grain size (El-Kadry *et al.* 1995). The increase in the grain size may be due to the coalescence of small crystals. The adatom mobility also increases as the substrate temperature increases which also results in the grain size and crystallinity of the films (Venkatachalam *et al.* 2006).



100 Transmittance, T and Reflectance, R (%) Transmittance 80 Room temp 60 100°C 150°C 40 200°C 20 30 90 12 15 18 21 24 27 0 0 00 00 00 00 00 Wavelength, λ (nm)

Fig. 3. Dependence of dislocation density and strain of ZnSe thin films having different substrate temperatures.

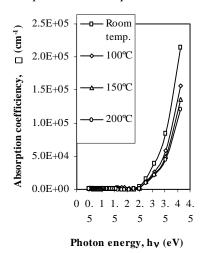
Fig. 4. Dependence of optical transmittance and reflectance on wavelength for ZnSe thin films having different substrate temperatures.

Fig. 3 shows the dislocation densities and strain decrease as the substrate temperature increases. Similar result for ZnSe thin films has been reported (Pal *et al.*1995). Since the dislocation density and strain are the manifestation of dislocation network in the films, the decrease in the strain and dislocation density indicates the formation of higher quality films at higher substrate temperatures. When the substrate temperature increases the line width narrows due to the increase in grain size.

The transmittance and reflectance spectra of ZnSe thin films deposited at different substrate temperatures having 300 nm thicknesses are shown in Fig. 4. The films demonstrate more than 80% transmittance at wavelengths longer than 550 nm, which confirms reasonably well (Ennaoui *et al.* 2003). Transmittance decreases sharply to almost zero bellow 550 nm that is due to the strong absorbance of the films in this region. The films deposited at lower substrate temperatures exhibited slightly less transmittance in the visible region.

The less transmittance observed at higher and lower substrate temperature could be due to the deviation of composition from stoichiometric structure of the films. The lower crystallinity of ZnSe layers deposited at such temperature may also be the reason for lower transmittance. The improved characteristics of the film at 200°C substrate temperature is also confirmed from absorption coefficient spectra, since it is observed that the spectrum for film at 200°C substrate temperature has more distinct absorption edge than those of other films. The reflectance spectra show the interference pattern with distinct peaks and valleys. It has also been found that the reflectance is small in the near infrared and visible region. The overall reflectance of the film increases with the increase in the substrate temperature.

The dependence of absorption coefficient on the photon energy for ZnSe thin films has been shown in fig. 5. These spectra reveal that the films show the significant absorption coefficient in the UV and visible regions. The values of absorption coefficient decrease with the increase in the substrate temperatures. This departure is caused owing to the presence of thermal lattice vibrations and imperfections. The increase in the substrate temperature produces a significant effect on the energy of the absorption edge. It shifts the absorption edge to the longer wavelength, which must be attributed to the growth of crystal grains and consequent decrease of quantum size effect (Kathalingam *et al.* 2007).



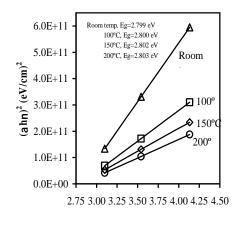


Fig. 5. Dependence of absorption coefficients on photon energy for ZnSe thin films having different substrate temperatures.

Fig. 6. Variation of $(\alpha h v)^2$ versus photon energy for ZnSe thin films deposited at different substrate temperatures.

Photon energy, hv (eV)

The analysis of the absorption coefficient in the photon energy range $2.48 \le hv \le 4.133$ eV follows a relation for an allowed direct band gap energy (Patel and Kapale 1987), described by

$$\alpha = \frac{A}{h\nu} \left[h\nu - E_g \right]^{1/2} \tag{1}$$

where, E_g is the band gap energy and A is a parameter that depends on the probability of transition and the refractive index of the material. The band gap energies and the values of A were found out from the plot of $(\alpha h v)^2$ versus h v as shown in figure 6. Extrapolation of the linear portion of the curve to $(\alpha h v)^2$ =0 gives the optical band gap and is found to be in the range from 2.799 to 2.803 eV for different substrate temperatures which are in good agreement with the literature (Kumar *et al.* 2007). The calculated optical parameters are summarized in table 2.

Table 2. Optical parameters of ZnSe thin films with varying substrate temperatures.

Substrate temperature (°C)	Eg (eV)	A (cm ⁻¹ eV ^{1/2})
Room temp.	2.799	6.68×10^5
100	2.800	4.83×10^5
150	2.802	4.19×10^{5}
200	2.803	3.75×10^{5}

The band gap energy (Eg) value is found to increase with the increase in the substrate temperature and the films are found to have direct allowed transition. The obtained Eg values are plotted as a function of substrate temperature in figure 7. The band gap energy increases slightly from 2.799 to 2.803 eV as the substrate temperature increases from room temperature to 200°C. This might be related to the existence of high density of mid band gap levels that could give rise to the band tailing in polycrystalline materials.

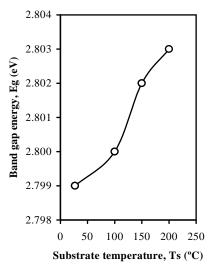


Fig. 7. Dependence of optical band gap energy with substrate temperature for ZnSe thin films.

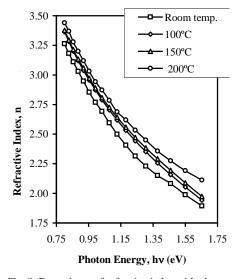


Fig. 8. Dependence of refractive index with photon energy for ZnSe thin films having different substrate temperatures.

Fig. 8 shows the variations in the refractive index with photon energy for ZnSe thin Films having different substrate temperatures. It is observed from the figure that the refractive indices of the films decrease with increasing photon energy. It may also be observed that the values of refractive index deposited at room temperature were found to be less than those films deposed at higher substrate temperatures, which is in good agreement with the literature (Pal *et al.* 1995). Such a difference may be due to the larger grain size and lower strain in the films deposited at higher temperatures.

CONCLUSIONS

The XRD reveals that the films have been found to be polycrystalline in nature having cubic zinc blende structure. The lattice parameter and grain size increases with the increase of substrate temperature. However the internal strain and dislocation density decreases with the increase of substrate temperature. The optical behavior exhibits direct band gap energies 2.799 to 2.803 eV. It may be asserted that thin films of ZnSe having adequate quality for photovoltaic devices may be produced by vacuum evaporation method.

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